

Optimizing transistor performance of percolating carbon nanotube networks

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In percolating networks of mixed metallic and semiconducting carbon nanotubes (CNTs), there is a tradeoff between high on-current (dense networks) and high on/off ratio (sparse networks). Experiments on transistors and Monte Carlo simulations were performed to determine the scaling behavior of device resistivity as a function of channel length (L) for CNT density (p) between 0.04 and 1.29 CNTs/ μm^2 in the on- and off-states (nanotube root mean square length of 5 μm). Optimized devices with field-effect mobility up to 50 $\text{cm}^2/\text{V s}$ at on/off ratio $>10^3$ were obtained at channel width $W=50 \mu\text{m}$ and $L>70 \mu\text{m}$ for $p=0.54\text{--}0.81$ CNTs/ μm^2 . © 2010 American Institute of Physics. [doi:10.1063/1.3469930]

Recently, random networks of carbon nanotubes (CNTs) as well as aligned arrays of CNTs have been demonstrated as potential active materials in large-area electronics applications.^{1–3} Although aligned arrays of CNTs can carry large channel current, they are plagued by low on/off ratios due to metallic CNTs directly shorting the electrodes.⁴ Metallic paths can be eliminated by electrical breakdown method^{3,5} but this has proved difficult to scale up for large area integrated electronics. In contrast, the effects of metallic CNTs in a random network can be mitigated by carefully controlling the CNT density (p) and device geometry such that the metallic fraction of the CNTs is below the percolation threshold (i.e., every conducting path contains at least one semiconducting CNT).^{1,6} This strategy contains an inherent tradeoff; high device current and field-effect mobility can be achieved by increasing the number of gate-tunable paths in the channel (i.e., by increasing CNT density). However, a very high CNT density results in reduced on/off ratio due to increased number of metallic CNTs. An optimized device (i.e., highest possible on-current at a given on/off ratio) has the total density of CNTs above the percolation threshold but the density of metallic CNTs below the percolation threshold. Although such high quality devices have been reported in the literature,^{1,2} a controlled and systematic experiment to determine the CNT density for optimum device performance is lacking. Here, we study the effect of device parameters (density of CNTs p and channel length L) on transport properties to obtain optimized device performance. The CNT density p was varied over two orders of magnitude by carefully controlling the chemical vapor deposition process. For each density, scaling behavior of device resistance as a function of channel length ($R \sim L^n$) was demonstrated and used to determine the overall percolation threshold and the percolation threshold of the metallic CNTs. We also performed Monte Carlo simulation of random networks of CNTs to confirm the observed percolative behavior of CNT thin films.

Field-effect mobility ($\sim 50 \text{ cm}^2/\text{V s}$) and on/off ratio ($\sim 4 \times 10^3$) of the optimized devices are comparable to the best devices reported in the literature.¹

CNT thin films were grown by chemical vapor deposition on 300 nm thick thermally oxidized Si substrates using Fe as the catalyst.^{7,8} As described elsewhere,⁸ the density of the CNT network was controlled in the range 0.04–1.29 CNTs/ μm^2 by varying the concentration of ferric nitrate catalyst. The density of CNTs was determined by counting the number of CNTs in field-emission scanning electron microscope (FE-SEM) images. CNTs were counted in $40 \times 60 \mu\text{m}^2$ rectangles (magnification=5000 \times) for sparse CNTs thin films ($p \leq 0.57$ CNTs/ μm^2) and in $12 \times 16 \mu\text{m}^2$ rectangles (magnification=20 000 \times) for denser CNT thin films ($p > 0.57$ CNTs/ μm^2). The root-mean square (rms) length and average length of the CNTs (l_{CNT}) was determined to be approximately 5 μm and 4.3 μm , respectively. Source-drain (S/D) electrodes (100 nm thick Au on 10 nm thick Ti) were deposited on the growth substrate by photolithography. Then, the CNT thin films were patterned in the device channel via another step of photolithography and reactive ion etching⁸ to obtain channel width $W=50 \mu\text{m}$ and channel lengths L varying from 5 to 100 μm in steps of 5 μm (20 devices in total). The resulting devices consist of a highly doped Si as global back gate, 300 nm thick SiO₂ dielectric layer, and top contact Ti/Au S/D electrodes.

Transport measurements of the devices were conducted in ambient conditions using a probe station (Cascade Microtech). Devices were characterized by computing device resistance, linear effective field-effect mobility (the field effect mobility assuming a homogeneous semiconductor film in the channel), and on/off ratio. We calculated the device resistance in the ON-state (R_{on} at $V_g = -30$ V) and the OFF-state (R_{off} at $V_g = 30$ V) from the inverse slope of the I_d - V_d characteristics in the linear regime $-1 \text{ V} < V_d < 1 \text{ V}$; V_g is gate voltage, V_d is drain voltage, and I_d is drain current. We studied the transport behavior as a function of L for ten different CNT densities in the range $p=0.04\text{--}1.29$ CNTs/ μm^2 . A total of 170 devices were measured (20 devices for each of

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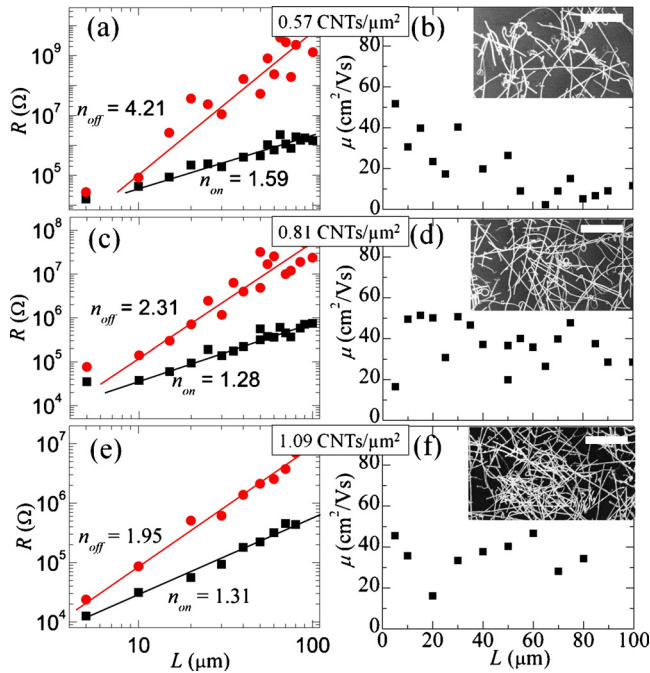


FIG. 1. (Color online) (a), (c), and (e) Device resistance (R) in ON-state (squares) and OFF-state (circles) vs channel length (L) as CNT network density (shown in the box in the center) is increased from top to bottom. n_{on} and n_{off} are scaling exponents ($R \sim L^n$) in ON- and OFF-state. Solid lines are best-fits to the experimental data from which n_{on} and n_{off} are extracted. (b), (d), and (f) Field-effect mobility (μ) vs L for different densities of CNTs. Insets in (b), (d), and (f) show FE-SEM images of corresponding CNT densities. Scale bars are $2 \mu\text{m}$ long.

CNT densities $p=0.04, 0.16, 0.23, 0.34, 0.57, 0.63$; and $0.81 \text{ CNTs}/\mu\text{m}^2$, and ten devices for each of CNT densities $p=1.09, 1.15$, and $1.29 \text{ CNTs}/\mu\text{m}^2$.

The transport properties of CNT thin film transistors (TFTs) have been interpreted within the framework of percolation theory of a random network of conducting sticks.^{9,10} Though scaling laws in percolation theory are valid for infinite sized systems, they offer approximate solution of the finite devices considered here. For $W \gg L$ and $L \gg l_{CNT}$, the resistance (R_s) of a conducting stick thin film is expected to follow a power law relation with channel length L .^{2,11} At the percolation threshold, R_s varies as $\sim L^{1.8}$, and at a density much above the percolation threshold, R_s varies linearly with L (approaching a homogeneous film).^{2,9-11} In between these two extremes, the exact form of the power law depends on the stick density. Factors which might affect the form of the power law observed experimentally include heterogeneity in the electronic type of CNTs (1/3 metallic and 2/3 semiconducting), CNT-to-CNT contact resistance,¹² twisted CNTs instead of straight sticks assumed in the model, and variations in CNT length.

Device resistance (R) is plotted against L as a log-log plot to extract the scaling exponent as illustrated in Fig. 1. In the ON-state, all the CNTs conduct, therefore the ON-state exponent, n_{on} ($R_{on} = L^{n_{on}}$) corresponds to the total density of CNTs. In the OFF-state, only metallic CNTs carry current, therefore n_{off} ($R_{off} = L^{n_{off}}$) is expected to give information about the density of metallic CNTs. For the smallest CNT density $p=0.04 \text{ CNTs}/\mu\text{m}^2$, only the smallest channel length device ($L=5 \mu\text{m}$) showed a conducting channel with $R \sim 100 \text{ k}\Omega$ and on/off ratio ~ 2 . In this case, the CNT thin film is too sparse to form a connected network and the con-

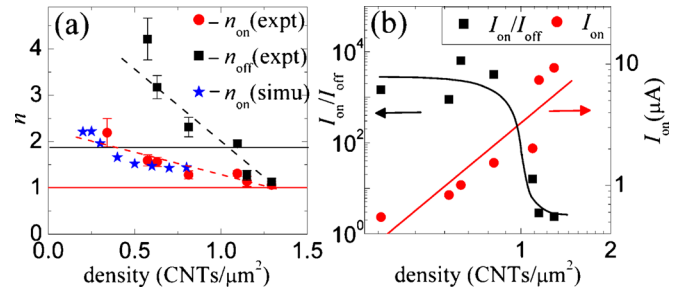


FIG. 2. (Color online) (a) Resistance power law exponents in ON- and OFF-states are plotted as a function of CNT density. Upper horizontal line at $n=1.8$ shows the percolation threshold, whereas lower horizontal line at $n=1$ represents the limit for a homogeneous film. The stars represent simulated ON-state exponents (see Fig. 3). Slanted dashed lines are to aid the eye. (b) On/off ratio and ON-state current are plotted as a function of CNT density (p) for a device with $L=100 \mu\text{m}$ and $W=50 \mu\text{m}$. Curved line is to aid the eye. Slanted line is a power law fit to ON-current ($I_{on} \sim p^{2.04 \pm 0.45}$).

ducting channel is formed by CNTs directly shorting the S/D electrodes. At higher CNT density, bigger clusters of connected CNTs result in more and more devices with conducting channels. For example, at $p=0.16$ and $0.23 \text{ CNTs}/\mu\text{m}^2$, conducting devices were obtained up to $L=10 \mu\text{m}$ and $25 \mu\text{m}$, respectively. At $p=0.34 \text{ CNTs}/\mu\text{m}^2$, all the devices up to $L=100 \mu\text{m}$ showed conducting channels with $n_{on}=2.19$, which suggests that density of CNTs is approaching the percolation threshold. For $p=0.57 \text{ CNTs}/\mu\text{m}^2$, R and linear field-effect mobility (μ) is plotted against L in Figs. 1(a) and 1(b), respectively. At this density, CNT TFTs showed field-effect mobility between $1-50 \text{ cm}^2/\text{V s}$ and on/off ratio in range $2-10^4$. In this case, the ON-state exponent $n_{on}=1.59$ and the OFF-state exponent $n_{off}=4.21$.

Figure 2(a) summarizes the extracted exponents n_{on} and n_{off} as a function of CNT density, and Fig. 2(b) shows the behavior of the on/off ratio ($I_{on}/I_{off} = R_{off}/R_{on}$) and the ON-state current I_{on} ($V_d = -1 \text{ V}$) as a function of CNT density for $L=100 \mu\text{m}$. Both the exponents n_{on} and n_{off} decrease with an increase in CNT density [see also Figs. 1(a), 1(c), and 1(e)]. The on/off ratio ($I_{on}/I_{off} = R_{off}/R_{on}$) of the devices increases with L [Fig. 1(a)] whereas field-effect mobility decreases with L [see Fig. 1(b)]. However, the dependence of the on/off ratio and field-effect mobility on L becomes weaker as CNT density is increased [Figs. 1(e) and 1(f)]. The observation of exponents n_{off} larger than 1.8 suggest that only the metallic CNT density is below the percolation threshold, without showing the extent of percolation in the network. In addition, the scaling behavior in the OFF-state resistance could be affected by the nonperfect turn-off of some semiconducting CNTs and the presence of a few long CNTs in the channel.

The average field-effect mobility and the ON-state current both increase with CNT density [Fig. 2(b)]. In contrast, the mean on/off ratio of the devices decreases with increasing network density [Figs. 1(a), 1(c), 1(e), and 2(b)], even though the on/off ratio of the devices with longer channel lengths ($L > 70 \mu\text{m}$) remains higher than 10^3 for $p \leq 0.81 \text{ CNTs}/\mu\text{m}^2$. This means metallic CNT clusters are not large enough to bridge long channels ($L > 70 \mu\text{m}$) for $p \leq 0.81 \text{ CNTs}/\mu\text{m}^2$ and the on/off ratio of the device remains in between 10^3 and 10^4 , whereas device current increases steadily with CNT density [see Fig. 2(b)]. At $p=1.09 \text{ CNTs}/\mu\text{m}^2$, the on/off ratio suddenly drops to the

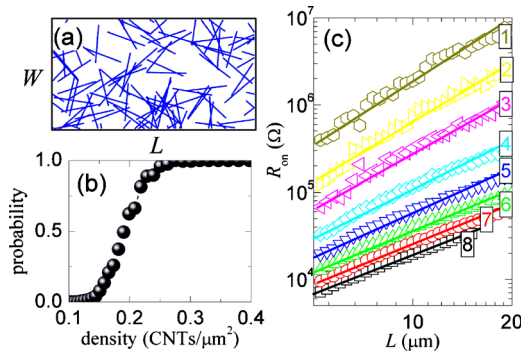


FIG. 3. (Color online) (a) A two-dimensional CNT network generated using Monte Carlo simulations. (b) Probability of connection between source and drain is plotted as a function of CNT density for $W=25 \mu\text{m}$, $L=35 \mu\text{m}$, and $l_{\text{CNT}}=5 \mu\text{m}$. The percolation threshold is defined as probability=0.99. (c) Simulated ON-state resistance of CNT devices is plotted against L for eight different CNT densities varying from 0.2 (no. 1) to 0.8 $\text{CNTs}/\mu\text{m}^2$ (no. 8).

range 4–40 and the OFF-state exponent becomes $n_{\text{off}}=1.95$ [Figs. 1(e) and 2(b)]. This suggests that density of metallic CNTs in the network has become close to the percolation threshold. For $p > 1.09 \text{ CNTs}/\mu\text{m}^2$, on/off ratio decreases further and both the exponents n_{on} and n_{off} approach 1 [Fig. 2(a)]. However, device current continues to increase with CNT density and follows a power law behavior [$I_{\text{on}} \sim p^{2.04 \pm 0.45}$, Fig. 2(b)]. The previously reported power law exponent for solution-processed CNTs (1.65 in Ref. 6) is within the error in the exponent from our as-grown CNTs. At the highest CNT density $p=1.29 \text{ CNTs}/\mu\text{m}^2$, the field-effect mobility of the devices is observed to be between 50 and $190 \text{ cm}^2/\text{V s}$ and the on/off ratio between 2 to 6.

Monte Carlo simulations were also performed to quantify the scaling behavior of the device resistance (ON-state), following the method described previously in Ref. 13. Random networks of $5 \mu\text{m}$ long CNTs (rms length of CNTs in experiments¹⁴) were generated in an area of fixed width and variable channel length, as shown in Fig. 3(a). It was observed that the scaling behavior in these device geometries becomes almost independent of W for $W > 15 \mu\text{m}$ ($W \gg l_{\text{CNT}}$).¹³ Therefore, W was fixed at $25 \mu\text{m}$ due to computational limitations. First, the percolation threshold was determined by calculating the connection probability (defined as the ratio of devices with connected S/D electrodes to total number of simulated devices) as a function of CNT density. An ensemble of 200 devices was simulated for each CNT density in a $W=25 \mu\text{m} \times L=35 \mu\text{m}$ channel area. The percolation threshold, which we define as the density at which the connection probability is 0.99 was found to be $0.28 \text{ CNTs}/\mu\text{m}^2$, [Fig. 3(b)]. This simulated percolation threshold is close to the minimum CNT density ($p=0.34 \text{ CNTs}/\mu\text{m}^2$) where all the devices (up to $L=100 \mu\text{m}$) showed a connected channel. Next, ON-state resistances (R) of the simulated networks were calculated based on the resistance values assigned to CNTs and their junctions.¹³ Figure 3(c) shows R as a function of L for CNT densities varying from 0.2 to $0.8 \text{ CNTs}/\mu\text{m}^2$ (numbered 1

to 8). Solid lines are least-squares fits to extract scaling exponents that compare favorably with experimental ON-state exponents (n_{on}), as seen in Fig. 2(a). In the density range of $p=0.34\text{--}0.81 \text{ CNTs}/\mu\text{m}^2$, $n_{\text{on}} < 1.8$ and $n_{\text{off}} > 1.8$; the overall CNT density is above the percolation threshold, whereas the density of metallic CNTs is below the percolation threshold. Devices with optimum performance (field-effect mobility $\sim 10\text{--}50 \text{ cm}^2/\text{V s}$, on/off ratio $> 10^3$) are obtained at $L > 70 \mu\text{m}$ and $p=0.34\text{--}0.81 \text{ CNTs}/\mu\text{m}^2$. The highest field-effect mobility, for the devices with on/off ratio $> 10^3$, is obtained at $p=0.81 \text{ CNTs}/\mu\text{m}^2$, which is the highest density CNTs where the density of metallic CNTs is still below the percolation threshold. The field-effect mobility (up to $50 \text{ cm}^2/\text{V s}$) and on/off ratio (up to 4×10^3) of these devices is comparable to one of the highest quality devices reported in literature¹, where the on/off ratio was improved by limiting metallic paths via lateral confinement of CNTs in long and narrow stripes. Here, comparable device performance was achieved in as-grown CNTs without utilizing lateral confinement¹ or electrical breakdown⁵ to minimize the effect of metallic CNTs. Presumably such additional processing or using sorted semiconducting CNTs¹⁵ could further enhance the device performance.

In conclusion, we performed controlled and systematic experiments and simulations to obtain the range of device parameters for optimized performance of CNT TFTs. Using this approach, CNT TFTs with field-effect mobility between 5 and $50 \text{ cm}^2/\text{V s}$ and on/off ratio $> 10^3$ were obtained at $W=50 \mu\text{m}$, $L > 70 \mu\text{m}$, and $p=0.54\text{--}0.81 \text{ CNTs}/\mu\text{m}^2$.

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¹Q. Cao, H. S. Kim, N. Pimparkar, J. P. Kulkarni, C. J. Wang, M. Shim, K. Roy, M. A. Alam, and J. A. Rogers, *Nature (London)* **454**, 495 (2008).

²E. S. Snow, J. P. Novak, P. M. Campbell, and D. Park, *Appl. Phys. Lett.* **82**, 2145 (2003).

³S. J. Kang, C. Kocabas, T. Ozel, M. Shim, N. Pimparkar, M. A. Alam, S. V. Rotkin, and J. A. Rogers, *Nat. Nanotechnol.* **2**, 230 (2007).

⁴C. Kocabas, N. Pimparkar, O. Yesilyurt, S. J. Kang, M. A. Alam, and J. A. Rogers, *Nano Lett.* **7**, 1195 (2007).

⁵K. Ryu, A. Badmaev, C. Wang, A. Lin, N. Patil, L. Gomez, A. Kumar, S. Mitra, H. S. P. Wong, and C. W. Zhou, *Nano Lett.* **9**, 189 (2009).

⁶H. E. Unalan, G. Fanchini, A. Kanwal, A. Du Pasquier, and M. Chhowalla, *Nano Lett.* **6**, 677 (2006).

⁷V. K. Sangwan, V. W. Ballarotto, M. S. Fuhrer, and E. D. Williams, *Appl. Phys. Lett.* **93**, 113112 (2008).

⁸V. K. Sangwan, V. W. Ballarotto, D. R. Hines, M. S. Fuhrer, and E. D. Williams, *Solid-State Electron.* **54**, 1204 (2010).

⁹G. E. Pike and C. H. Seager, *Phys. Rev. B* **10**, 1421 (1974).

¹⁰D. J. Frank and C. J. Lobb, *Phys. Rev. B* **37**, 302 (1988).

¹¹S. Kumar, J. Y. Murthy, and M. A. Alam, *Phys. Rev. Lett.* **95**, 066802 (2005).

¹²M. A. Topinka, M. W. Rowell, D. Goldhaber-Gordon, M. D. McGehee, D. S. Hecht, and G. Gruner, *Nano Lett.* **9**, 1866 (2009).

¹³A. Behnam, G. Bosman, and A. Ural, *Phys. Rev. B* **78**, 085431 (2008).

¹⁴J. Hicks, A. Behnam, and A. Ural, *Phys. Rev. E* **79**, 012102 (2009).

¹⁵M. S. Arnold, A. A. Green, J. F. Hulvat, S. I. Stupp, and M. C. Hersam, *Nat. Nanotechnol.* **1**, 60 (2006).